Photocatalytic Degradation of RhB by Fluorinated Bi$_2$WO$_6$ and Distributions of the Intermediate Products

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Received October 2, 2007. Revised manuscript received January 6, 2008. Accepted January 7, 2008.

Fluorinated Bi$_2$WO$_6$ catalyst was synthesized by a simple hydrothermal process. The effects of fluoride doping on crystal structure, optical property, photoinduced hydrophilicity, surface acidity, and photocatalytic activity of the as-prepared sample were observed in detail. Fluorinated Bi$_2$WO$_6$ presented the enhanced photocactivity for the RhB degradation under the cumulative sunlight ($\lambda > 290$ nm), which could be a synergetic effect of the surface fluorination and the doping of crystal lattice. To get a better handle on the mechanistic details of this photocatalytic system, the photodegradation process of RhB was examined. In the fluorinated Bi$_2$WO$_6$ system, five intermediates, namely, N,N-diethyl-N-ethylrhodamine, N,N-diethylrhodamine, N-ethyl-N-ethylrhodamine, and rhodamine were thus identified, whereas the first three intermediates could only be identified in the case of the Bi$_2$WO$_6$ system. This result indicated that more RhB molecules were degraded via the deethylation process in the fluorinated Bi$_2$WO$_6$ system. It was proposed that the F$^{-}$-containing function on the catalyst surface could serve as an electron-trapping site and enhance interfacial electron-transfer rates by tightly holding trapped electrons. On the basis of the experimental results, a photocatalytic mechanism was discussed in detail.

1. Introduction

Semiconductor photocatalytic processes have been widely applied as techniques of destruction of organic pollutants in wastewater and effluents. To date, the major research on photocatalytic oxidation techniques is focus on TiO$_2$, as it is of low cost, is environmentally friendly, and has high stability (1). Very recently, some tungstates such as Bi$_2$WO$_6$ and ZnWO$_4$ have also presented high photocatalytic activities for the pollutant degradation (2–8). Their unique combination of physical and chemical properties, in terms of molecular and electronic versatility, reactivity, and stability, make us have reason to believe they may be a promising class of photocatalysts for use in degrading organic pollutants.

A major limitation of achieving high photocatalytic efficiency in semiconductor systems is the quick recombination of charge carriers. Recombination, which has faster kinetics than surface redox reactions, is a major drawback as it reduces the quantum efficiency of photocatalysis. Therefore, ways to minimize the recombination rate are important if we are interested in maximizing the photocatalysis efficiency. For this purpose, fluorinated TiO$_2$ has been investigated in relation to doping (TiO$_2$-F$_x$) or surface complexation (F$^{-}$-TiO$_2$) (9–13). Hattori et al. reported that fluoride doping improves the crystallinity of anatase and the photocatalytic reactivity. In addition, TiO$_2$-F$_x$ has fewer anion vacancies with a lower density of midgap states and is more stable against photocorrosion (9). Yu et al. proposed that the doping F$^{-}$ ions convert Ti$^{4+}$ to Ti$^{3+}$ by charge compensation and that the presence of a certain amount of Ti$^{3+}$ reduces the electron–hole recombination rate and thus enhances the photocatalytic activity (10). Recent study confirmed that F-doping in TiO$_2$ can also induce a visible photocatalytic activity by the creation of oxygen vacancies (11). On the other hand, surface fluorination of TiO$_2$ is a simple ligand exchange between F$^{-}$ ions and surface hydroxyl groups on TiO$_2$. It was reported that the surface fluorination of TiO$_2$ improves the photocatalytic oxidation rate of phenol and tetramethylammonium at a specific pH range (12). Since the surface fluorides themselves should not be reactive with valence band (VB) holes [E$^0$(F$^-$/F$^-$) = 3.6 V vs NHE], the higher photocatalytic oxidation rate in the F$^{-}$-TiO$_2$ suspension has been ascribed to the enhanced generation of mobile free OH radicals. Park and Choi reported that surface fluorination of TiO$_2$ changes not only the photodegradation rate of the pollutant but also the mechanistic pathways of the pollutant degradation, as well as the intermediates and product distribution subsequently (13).

More recently, a few F-doped non-TiO$_2$ catalysts also exhibited a similar enhancement of photocatalytic activities. F-doped SrTiO$_3$ shows about three times the photocatalytic activity compared with that of undoped SrTiO$_3$ (14). The enhancement of photocatalytic activity was mainly ascribable to the formation of oxygen vacancies and increase of effective electron mobility. Fluorine interstitially doped ZnWO$_4$ has been reported recently by our group (8). The doping fluorine ions could increase the coordination sphere around the W atom in a WO$_6$ octahedron and cause the distortion of the WO$_6$ octahedron in a ZnWO$_4$ crystal, resulting in an increased transfer rate of photogenerated electrons to the photocatalyst surface and the enhanced photoactivity for rhodamine B (RhB) degradation. On the basis of these results, it could be inferred that the introduction of fluorine, irrespective of the approaches of fluorination, really enhanced the activity of photocatalysts.

We report herein the photocatalytic performances of fluorinated Bi$_2$WO$_6$, prepared by a hydrothermal process. Bulk and surface characterizations of the resulting powders were carried out by means of X-ray diffraction (XRD), determination of Brunauer–Emmett–Teller (BET) specific surface areas, porosity measurements, and diffuse reflectance spectroscopy (DRS) techniques. A RhB dye was used as a model to examine the activity of the photocatalyst. To provide an overall understanding of the reaction pathway(s), high-performance liquid chromatography (HPLC), liquid chromatography/mass spectrometry (LC-MS), and gas chromatography/mass spectrometry (GC-MS) were exploited to identify the N-dealkylated intermediates and the final products of RhB from ring cleavage during the photodegradation. This work may provide new insights and under-
standing on the photocatalytic mechanisms of the fluorination of non-TiO₂ catalysts.

2. Experimental Section

2.1. Materials and Sample Preparation. Fluorinated Bi₂WO₆ samples were prepared by a two-step hydrothermal process. The starting materials of WO₃, Na₂O, and NaF were mixed and calcined at 1100 °C. The final Na₂WO₄ containing F⁻ ions was obtained from XRD patterns (see Supporting Information, Figure S1). In what followed, the value of RF (a) was used to describe the molar ratio of NaF to Na₂O; these were 0, 0.1, 0.2, 0.4, and 0.6 nominal ratios. The calcined sample and Bi(NO₃)₃ (the molar ratio of 1:1) were used to synthesize fluorinated Bi₂WO₆ by a hydrothermal process as described in our previous report (3). The resulting products were collected by filtration and then were washed by deionized water several times until no F⁻ ions were left in the solution as tested. The samples were then dried at 80 °C for 4 h for characterization. TEM and HRTEM images of the typical sample (RF = 0.4) show that the crystals are sheet-shaped (see Supporting Information, Figure S2 and Figure S3). All chemicals used were analytical grade reagents without further purification. Deionized water was used throughout this study.

2.2. Characterization. X-ray diffraction patterns of the powders were recorded at room temperature by a Bruker D8 Advance X-ray diffractometer using Cu Kα radiation and a 2θ scan rate of 2 min⁻¹. The diffuse reflectance absorption spectra of the samples were recorded on a UV–visible spectrophotometer (Hitachi UV-3100) equipped with an integrated sphere attachment. BET surface area measurements were performed by a Micromeritics ASAP 2010 V5.02H surface area analyzer. The nitrogen adsorption and desorption isotherms were measured at −196 °C after degassing the samples on a Sorptomatic 1900 Carlo Erba instrument. A photoinduced change of water contact angle (CA) was measured by a commercial contact angle meter (DropMaster 300, Kyowa Interface Science, Japan) at ambient condition, and UV irradiation was produced by a black lamp bulb. To observe the surface acidity of the as-prepared sample, temperature-programmed desorption (TPD) of the preadsorbed pyridine was performed in a vacuum chamber equipped with a Nicolet 470 FTIR spectrometer with a MCT/B infrared detector.

2.3. Photocatalytic Experiments. The photocatalytic activities of the as-prepared samples for the degradation of RhB in solution were tested under the simulative sunlight. A 500 W xenon lamp (λ > 290 nm, CHF-XM-500 W, the Trusttech. Co. Ltd., Beijing) was used as light source, and the average light intensity was 50 mW·cm⁻². In a 100 mL suspension containing 2 × 10⁻⁵ M RhB, 100 mg of the powder samples was dispersed in a rectangular vessel. Prior to irradiation, the suspensions were magnetically stirred in the dark for 0.5 h to ensure the establishment of an adsorption–desorption equilibrium. The suspensions were kept under constant air-equilibrated conditions before and during the irradiation. Between intervals, the solutions were centrifuged to remove the particles and filtrates were analyzed by recording the variations in the absorbance band (553 nm) in the UV–visible spectra of RhB using a Hitachi U-3010 UV–vis spectrometer.

2.4. Analyses of Intermediate Products. The N-dealkylated intermediates and the original dye were detected by an HPLC technique. The HPLC system consisted of a Dionex P580 pump, a UVD 340S diode array detector, and an isothermal ODS-3 C18 reverse column (5 μm, 250 × 4.6 mm²). Five intermediates, namely, N,N-diethyl-N'-ethylrhodamine (DER), N,N-diethylrhodamine (DR), N-ethyl-N'-ethylrhodamine (EER), N-ethylrhodamine (ER), and rhodamine (R) were thus identified. LC-MS analyses were carried out using a commercial Thermo Finnigan (San Jose CA) LCQ Advantage ion trap mass spectrometer equipped with an ESI ion source. GC-MS analyses were carried out on a Finnigan Trace DSO Ultra instrument equipped with a DB-5 MS capillary column (30 m × 0.25 mm).

3. Results and Discussion

3.1. Catalyst Characterization. X-ray diffraction patterns were used to investigate the changes of the phase structures of the as-prepared samples. Figure 1 shows XRD patterns of the samples prepared by the hydrothermal process. All of the as-prepared samples appeared to be phase-pure Bi₂WO₆ [JCPDS No. 73-1128]. Further observation showed that the peak intensities of Bi₂WO₆ became steadily weaker with increasing RF, and the diffraction peaks of Bi₂WO₆ were wider, indicating the reducing of crystallization and formation of smaller Bi₂WO₆ crystallites. The presence of F⁻ ions could play an important role in the formation of Bi₂WO₆ crystals. It was well-known that the phase formation of Bi₂WO₆ crystal occurred due to the rearrangement of WO₆ octahedra (3, 5). The crystallization process of the Bi₂WO₆ phase from the amorphous precursors was postulated to initiate through face-sharing polycondensation of these octahedrons. According to the surface acidic/basic properties, the surface Bi–OH group in the Bi₂WO₆ crystal should be at least partially protonated to give BiOH₂⁺. Consequently, the free F⁻ ions liberated in the solution would interact with the small crystal. Nucleophilic substitution was favored due to the high electronegativity of F⁻ ions, which in turn reduced the chemical reactivity of the precursor. In the process of the crystallization, F⁻ ions were involved, resulting in the more vacancies in the Bi₂WO₆ crystal. Although more work is needed to illustrate clearly these rather complex interactions, it is believed that the interaction of these anionic species with the surface Bi³⁺ ion prohibited the enlargement of the crystal nuclei.

Figure 2 shows the nitrogen adsorption and desorption isotherms of the as-prepared samples with varying RF. The isotherm corresponding to the sample obtained without fluoride was of type IV (BDDT classification) with two capillary condensation steps, implying a bimodal pore size distribution in the mesoporous and macroporous regions. This bimodal mesopore size distribution resulted from two different aggregates in the powders. The hysteresis loop in the lower relative pressure range (0.4 < P/P₀) was related to finer intraaggregated pores formed between

![Figure 1](image-url)
intraagglomerated primary particles, and that in the higher relative pressure range \((0.8 < P/P_0 < 1)\) was associated with larger interaggregated pores produced by interaggregated secondary particles \((1.5)\). The bimodal mesopore size distribution was further confirmed by the corresponding pore size distributions shown in the inset of Figure 2. The powder contained small mesopores of ca. 1.6 nm and larger mesopores with a maximum pore diameter of ca. 2.6 nm. The presence of fluoride in the synthetic system exerted a significant influence on the pore structures of the obtained products. With increasing \(R_F\), the isotherms showed higher adsorption at high relative pressures, indicating the formation of the mesopores and/or an increasing pore volume. The two separate hysteresis loops gradually joined together as one loop, implying that the pore size distributions of the intraagglomerated and interagglomerated pores tended to overlap, as confirmed in the inset of Figure 2. Furthermore, the pore volumes of intraagglomerated mesopores were negligible compared with those of the interagglomerated ones, based on their corresponding integral areas. The BET surface areas of Bi\(_2\)WO\(_6\) increased from 0.2 to 0.6. We concluded that the difference in the adsorption edges could arise from the fluorination of Bi\(_2\)WO\(_6\). For a crystalline semiconductor, it is shown in the inset of Figure 3 that the optical absorption near the band edge follows the Kubelka–Munk functions against the photon energy, as shown in the inset of Figure 3. The band gap of the sample was estimated to be 2.71, 2.71, 2.70, 2.55, and 2.50 eV from the onset of the absorption edge, corresponding to \(R_F = 0\), 0.1, 0.2, 0.4, and 0.6, respectively. The color of the sample \((R_F = 0, 0.1, 0.2)\) was pale-yellow, whereas the sample was yellowish in color when \(R_F = 4\) or 0.6, as predicted from their photoabsorption spectra.

To study the surface properties of the as-prepared samples, the variation of CA under weak UV light irradiation was observed, and the results are shown in Figure S4. Before UV light irradiation, the original CA for the Bi\(_2\)WO\(_6\) sample was about 39°. With \(R_F\) increasing, CA for the fluorinated Bi\(_2\)WO\(_6\) decreased to 12° \((R_F = 0.6)\), suggesting that the fluorination greatly improved the duration of hydrophilicity of the material (see Supporting Information, Figure S5). With UV light irradiation, all the samples showed photoinduced hydrophilicity, whereas CA decreased quicker on the sample with smaller \(R_F\) in most cases. Hashimoto et al. found that the photoinduced hydrophilicity was due to structural changes of the material surface, completely different from the conventional photocatalytic reaction. The key was the hydroxyl group’s reconstruction and that their amount increased on the sample surface under UV light irradiation \((16)\). In detail, the photogenerated holes were trapped by surface lattice oxygen atoms, resulting in a breaking of the bonds between oxygen and metal atoms. Subsequently, water molecules dissociatively absorbed at these sites (hydroxyl formation). So the bond strength between oxygen and metal atom was important to the photoinduced hydrophilicity. The fluorination can increase the hydroxyl amount of the surface in the aqueous solution. Meanwhile, because F\(^-\) ions have a much stronger affinity for electrons compared with O\(^2-\) ions, it is postulated that the electronic cloud could deviate from the original position in a Bi–O polyhedron containing fluoride, and the bonds between Bi and O could be weakened in F–Bi\(_2\)WO\(_6\). Therefore, plenty of oxygen vacancies are easily produced with the illumination of UV light in F–Bi\(_2\)WO\(_6\). This could undoubtedly improve the photoinduced hydrophilicity. In addition, the fluorinated process could probably increase defects, which is considered to be beneficial to the decrease in the water CA. This could be the reason that the original CA on F–Bi\(_2\)WO\(_6\) is smaller than that of Bi\(_2\)WO\(_6\). The CA experiment revealed that the fluorination remarkably affected the surface properties of Bi\(_2\)WO\(_6\) catalyst and could give rise to a great influence on the photochemical properties of Bi\(_2\)WO\(_6\) catalyst \((17)\).
The surface acidity of the as-prepared sample was also examined as a correlation between the surface acidity and photocatalytic activity of the photocatalysts. The adsorption of pyridine, monitored by FTIR, was chosen as a probe reaction to characterize the surface acidity. The results are shown in Figure 4. The peaks at the range of 1400–1650 cm⁻¹ were ascribed to the pyridine ring-stretching modes. Upon the interaction with a Brønsted acid, pyridine was protonated and absorbed at a specific IR wavelength around 1540–1545 cm⁻¹, as well as at several other specific wavelengths not discussed here. Following interaction with a Lewis acid site, pyridine would also form a coordinatively bonded complex with a specific IR absorption band centered at 1445 cm⁻¹ (10). Therefore, the bands at 1541 and 1440 cm⁻¹ observed for both samples could be assigned to the coordinated pyridine adsorbed on the Brønsted acid and Lewis acid surface sites, respectively. With the increase of temperature to 150 °C, the peaks at 1541 and 1440 cm⁻¹ on the Bi₂WO₆ sample disappeared swiftly. However, in the F–Bi₂WO₆ case, these bands disappeared completely only when the sample was treated at 200 °C. This result indicated that the fluorinated Bi₂WO₆ has stronger acid sites, and enhanced acidity of the surface, which could contribute to higher photocactivity.

### 3.2. Photodegradation of RhB and Identification of the Intermediates

The RhB photodegradation over the as-prepared samples under the simulated sunlight (λ > 290 nm) is shown in Figure S6. The photodegradation of RhB under the simulated sunlight without a catalyst was insignificant. The photodegradation of the sample for the RhB photodegradation was enhanced. With the exclusion of the sample (R₀ = 0.6), the as-prepared sample showed higher photoactivities for the RhB degradation than that of P25 (TiO₂). The maximum photocactivity was observed for the sample (R₀ = 0.4). When R₀ further increased to 0.6, the photocactivity of the sample instead decreased. Although it presented a surface area (51.3 m²/g) similar to that of the sample of R₀ = 0.4 (48.2 m²/g), their photocatalytic activity was fairly different. Therefore, it was inferred that the specific surface areas played a minor role in this photocatalytic process.

The temporal evolution of the spectral changes taking place during the photodegradation of RhB over Bi₂WO₆ (R₀ = 0) and F–Bi₂WO₆ (R₀ = 0.4) is shown in Figure 5, parts A and B, respectively. Tetraethyalted rhodamine shows a major absorption band at 553 nm. In both cases, the irradiation for 180 min, which proved more RhB molecules were degraded by the N-demethylation process in the F–Bi₂WO₆ system as compared to the Bi₂WO₆ system. To obtain more information about the highly polar and small molecular weight intermediates that could not be separated and identified by HPLC, the products were further analyzed by a GC-MS technique. The major final products in the presence of Bi₂WO₆ or F–Bi₂WO₆ detected are listed in Figure S9 (see Supporting Information). They were mainly formic acid, phthalic acid, and phthalic anhydride. Many N-containing compounds with small molecular weight were also observed, which indicated that N-containing fraction of the RhB molecule could first be attacked by the active
species, and the RhB molecules were degraded subsequently. The GC-MS analysis provided solid evidence for the cleavage of the conjugated xanthene structure of both RhB and the deethylated intermediates during the degradation reaction in the presence of two photocatalytic systems.

### 3.3. The Proposed Mechanism

On the basis of the DRS results, it could be speculated that F⁻ ions incorporated into the structure of Bi₂WO₆ prepared by a fluoride-mediated hydrothermal process. These F⁻ ions may be positioned in the crystal lattice, may be an interstitial impurity, or may be adsorbed on the surface.

X-ray diffraction patterns illustrated that the doping with fluorine did not result in the development of new crystal orientations or a change in preferential orientations. Hence, independent of the presence or absence of fluorine, the samples were the pure aurivillius Bi₂WO₆ phase. However, a careful comparison of the (113) diffraction peaks in the range of 2θ = 27.5–29.5°, as shown in Figure S10, the peak position of Bi₂WO₆ with different fluorine contents shifted slightly toward a higher 2θ value. The same results were also presented in the other diffraction peaks. According to Bragg’s law, \( d_{hkl} = \lambda / (2 \sin \theta) \), where \( d_{hkl} \) is the distance between crystal planes of \( (hkl) \), \( \lambda \) is the X-ray wavelength, and \( \theta \) is the diffraction angle of the crystal plane \( (hkl) \). The ionic radius of F⁻ (0.133 nm) is smaller than that of O²⁻ (0.140 nm); the decrease in lattice parameters \( (d_{hkl}) \) value should result in the increase in 2θ values. Therefore, the observed shift of the diffraction peaks toward higher angles could be due to the smaller lattice parameter expected for the substitution of O²⁻ by F⁻. Instead, F⁻ doping as the interstitial dopants could lead to an increase in lattice parameter due to repulsion between F⁻ anions, which has been observed from the fluorinated ZnWO₄ in our previous report (6). Accompanied by F⁻ doping, surface fluorination of Bi₂WO₆ occurred concurrently, which could be concluded from Figures 3 and 4. Therefore, the enhanced photoactivity of F⁻Bi₂WO₆ could be a synergetic effect of surface modification and bulk doping by F⁻ ion.

The Bi₂WO₆ formation was a typical hydrothermal ripening process: a highly supersaturated solution was adopted, and the amorphous fine particles acted as the precursor for the synthesis of crystallized Bi₂WO₆. In the beginning, the formation of tiny crystalline nuclei in a supersaturated medium occurred, and the crystal growth then followed. The larger particles grew at the cost of the small particles, according to the well-known Gibbs–Thomson law. This speculation has been proved by the TEM observation in our previous report (5). In the present condition, F⁻ ions were introduced from the precursor of fluorinated Na₂WO₄, which was synthesized from WO₃, Na₂O, and NaF by the high-temperature solid-state reaction. One fraction of F⁻ ions by the substitution of O²⁻ in the crystal lattice of Na₂WO₄ could

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**FIGURE 5.** UV–visible spectral changes of RhB \((2 \times 10^{-5} \text{ M}, 50 \text{ mL})\) over the catalysts, (A) \( R_F = 0 \) and (B) \( R_F = 0.4 \). Variations in the distribution of the intermediate products in the corresponding suspensions, (C) \( R_F = 0 \) and (D) \( R_F = 0.4 \). The concentrations of the \( N \)-dealkylated intermediates were determined by the corresponding peak areas gained from the HPLC technique.
be directly introduced to the resulting sample as the doping of the crystal lattice. Instead, the other fraction of F$^-$ ions could be liberated into the solution in the hydrothermal process. It is well-known that the aurivillius structure of Bi$_2$WO$_6$ consists of alternating fluoride-like [Bi$_2$O$_2$]$_{2+}$ layers and [WO$_4$]$^{2-}$ layers which comprise the corner-linked octahedral units. Due to the high electron density of F$^-$ ions, there is a strong repulsion between F$^-$ and [WO$_4$]$^{2-}$. Thus, the dissolution of F$^-$ ions into the interstitial lattice between [Bi$_2$O$_2$]$_{2+}$ layers and [WO$_4$]$^{2-}$ layers is limited. However, this fraction of F$^-$ ions could interact strongly with the surface of the Bi$_2$WO$_6$ crystal, which prohibited the enlargement of the crystal, and enhanced the surface acidity of the Bi$_2$WO$_6$ catalyst.

A few researchers have reported the improvement of photocatalytic activity by the fluorination of TiO$_2$ powders (9–13, 21). This high photocatalytic activity was ascribed to several beneficial effects produced by the F$^-$ ion doping in bulk TiO$_2$ crystal: enhancement of the crystallinity, creation of oxygen vacancies, or increase of active sites by conversion of some Ti$^{4+}$ to Ti$^{3+}$. As for the surfaces fluorination of TiO$_2$, the enhanced photoactivity of TiO$_2$ was commonly ascribed to a higher surface acidity, enhanced generation of mobile free OH radicals, and surface oxygen vacancies. In this case, the resulting Bi$_2$WO$_6$ sample prepared by a fluoride-mediated synthesis showed the pure aurivillius phase but a fall in the standard redox potential of Bi$^{3+}$/Bi$^{4+}$ being more negative than that of OH$^-$/OH$^-$ ($+1.59$) (3). This high photocatalytic activity of the as-prepared sample could partly explain the enhanced photoactivity. Following this general information, we assume that the enhanced photoactivity of F$^-$–Bi$_2$WO$_6$ could be a synergetic effect, including more oxygen defects, enhancement of surface acidity, and light absorbance.

In the present study, the fluorination of Bi$_2$WO$_6$ affected not only the rate but also the mechanistic pathways of the RhB degradation. In the Bi$_2$WO$_6$-mediated photocatalysis, the degradation of RhB was completely ascribed to a photochemical process involved by photoinduced holes. Furthermore, there could be two photochemical mechanisms, namely, a photocatalytic process and a photosensitized process. On one hand, Bi$_2$WO$_6$ was an active photocatalyst, and RhB could be degraded by the direct interaction with a strong oxidizing hole originating from the hybridization of the Bi6s and O2p orbitals. The RhB molecule was attacked and then degraded via the destruction of the conjugated structure. On the other hand, RhB dye could absorb the visible light, which was attributed to the ground state and the excited state of the dye. Therefore, RhB could be degraded by a photosensitized process. In this process, the degradation of RhB did not involve any VB hole; mineralization of dyes found their origins with the active oxygen radical species, O$_2^*$, and the radical cations, dye$^*$. The photosensitized degradation of RhB was commonly via the N-demethylation process (19). Compared to the photocatalytic process, the photosensitized process required the stronger interaction between the dye and the Bi$_2$WO$_6$ catalyst, which was beneficial to the transfer of photoinduced electron. The F-containing function on the fluorinated surface seemed to serve as an electron-trapping and enhanced interfacial electron-transfer rates by tightly holding trapped electrons. As a result, more RhB molecules were degraded via the N-demethylation process. It has been reported that the surface fluorination of TiO$_2$ by the replacement of OH by F$^-$ ions changed the surface interactions. Then, the rate of electron scavenging by O$_2$ and superoxide formation would be influenced (21).

**Acknowledgments**

This work is supported by the National Natural Science Foundation of China (Grants 20433010, 20571047, 20747002 and 20377008) and the National Program of China (Grant 2007CB613033).

**Supporting Information Available**

XRD patterns of F$^-$–Na$_2$WO$_4$, TEM, HRTEM, and CA, and XA patterns of F$^-$–Bi$_2$WO$_6$, photodegradation of RhB, HPLC, LC-MS, and GC–MS analysis of the intermediate products, and diffraction peak variation of F$^-$–Bi$_2$WO$_6$. Figures S1–S10. This material is available free of charge via the Internet at http://pubs.acs.org.

**Literature Cited**


ES702495W